

THERMOANALYTICAL AND SINGLE CRYSTAL GROWTH INVESTIGATIONS IN THE  
SYSTEM  $\text{Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  AND  $\text{Bi}_2\text{O}_3\text{-Nd}_2\text{O}_3\text{-P}_2\text{O}_5$

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ABSTRACT

The melting diagram  $\text{Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  has been studied by DTA and simultaneous DTA-high temperature microscopy in the composition range 80 to 92.7 mol%  $\text{Bi}_2\text{O}_3$ . A compound is found at the ratio  $\text{Bi}_{5.80}\text{PO}_{11.20}$  melting congruently at  $951^\circ\text{C}$ . By the Czochralski technique single crystals were grown of this compound either pure or doped with neodymium.

INTRODUCTION

Single crystals are advantageously grown from a melt of that composition with the melting point maximum. DTA and high temperature <sup>microscopy</sup> (HTM) are well suited to characterize the melting and freezing behavior in a binary or more complex system, thereby they provide a model of the transformations underlying the crystal growth process.

A compound was reported to exist at the composition  $\text{Bi}_5\text{PO}_{10}$  in the  $\text{Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  section of the Bi-P-O ternary [1], melting congruently at  $977^\circ\text{C}$ , and to be grown as single crystal [1,2]. Our attempts to prepare  $\text{Bi}_5\text{PO}_{10}$  single crystals as new laser host material failed. Independently of the preparation method we obtained a mixture that exhibited partial eutectic melting. Better knowledge of the relevant part of the pseudobinary  $\text{Bi}_2\text{O}_3\text{-P}_2\text{O}_5$  melting diagram was necessary as prerequisite for single crystal growth.

PHASE DIAGRAM STUDY

DTA and simultaneous DTA-HTM [3] were applied to mixtures of  $\text{BiPO}_4$  and  $\text{Bi}_2\text{O}_3$  to cover a concentration range 80 mol%  $\text{Bi}_2\text{O}_3$ , 20 mol%  $\text{P}_2\text{O}_5$ -92.7 mol%  $\text{Bi}_2\text{O}_3$ , 7.3 mol%  $\text{P}_2\text{O}_5$ .  $\text{BiPO}_4$  was prepared by sintering  $\text{Bi}_2\text{O}_3 + 2 \text{NH}_4\text{H}_2\text{PO}_4$ . Heating curves at  $\beta = 10 \text{ K min}^{-1}$  of samples  $\approx 60 \text{ mg}$  were evaluated, since the melts crystallized

after undercooling by 30 to 50 K. The eutectic effect was quantitatively estimated from the ratio of its peak height to the peak height of the liquidus effect. A calorimetric evaluation of the eutectic peak area gave worse results due to partial overlapping with the liquidus peak.

### RESULTS

Phase diagram results are shown on fig. 1. A liquidus temperature maximum appeared at a concentration 85.3 mol%  $\text{Bi}_2\text{O}_3$ , 14.7 mol%  $\text{P}_2\text{O}_5$  and  $951^\circ\text{C}$ . A eutectic was found at  $925^\circ\text{C}$  for  $< 85.3\%$   $\text{Bi}_2\text{O}_3$  and three peritectic transformations at  $932^\circ\text{C}$  for  $> 85.3$ , at  $918^\circ\text{C}$  for  $> 86.5$  and at  $865^\circ\text{C}$  for  $> 87.3$  mol%  $\text{Bi}_2\text{O}_3$ . Disappearance of the eutectic and peritectic effects suggested a narrow phase width corresponding to a formula  $\text{Bi}_{5.80+\delta}\text{P}_{11.20+3\delta/2}$  ( $-0.06 \leq \delta \leq +0.04$ ). The DTA diagrams of both a single crystal sample and powder of the composition

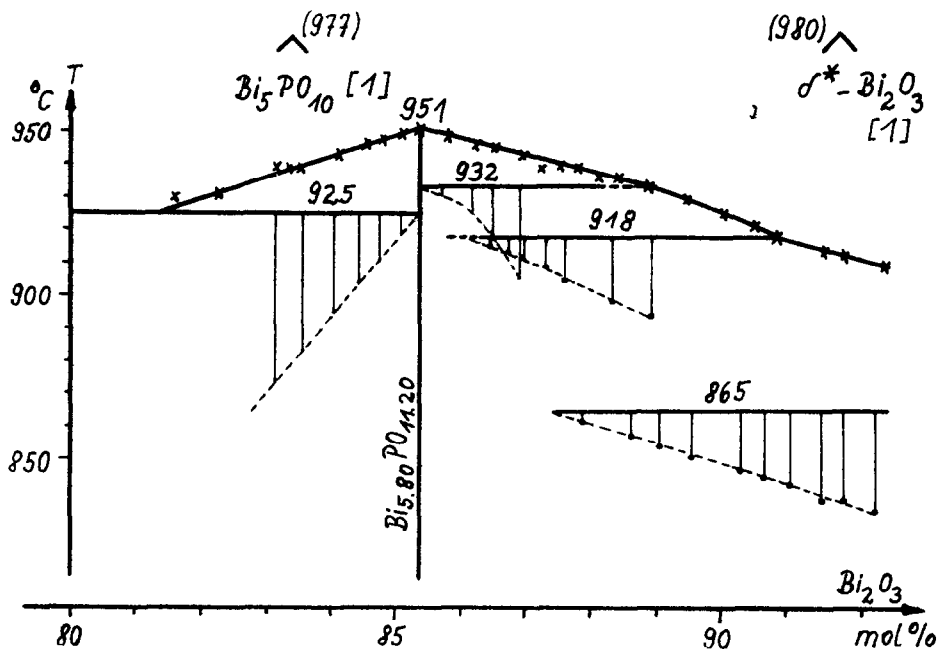


Fig. 1 melting diagram of the system  $\text{Bi}_2\text{O}_3$ - $\text{P}_2\text{O}_5$  between 80 and 92.5 mol%  $\text{Bi}_2\text{O}_3$ . Melting point maxima reported by Volkov et al. [1] are indicated.

$\text{Bi}_{5.80}\text{PO}_{11.20}$  were identical, only one melting peak is seen. No solid state transformation was detected between room and melting temperature. We assume that the non-stoichiometric compound was erroneously described by previous authors [1,2] as  $\text{Bi}_5\text{PO}_{10}$ . No evidence was found for another congruently melting phase between 91 and 92 mol%  $\text{Bi}_2\text{O}_3$ , assigned [1] to a non-stoichiometric  $\delta^*-\text{Bi}_2\text{O}_3$  phase.

#### SINGLE CRYSTAL GROWTH

In accordance with the thermoanalytical results single crystals could be easily grown from a melt  $\text{Bi}_{5.80}\text{PO}_{11.20}$  thus confirming this composition. Honey-yellow crystals were obtained by the Czochralski technique (RF heating, platinum crucible and after-heater, ambient atmosphere) with dimensions up to 15 mm diameter, 40 mm length.

#### NEODYMIUM DOPING

DTA and HTM experiments were performed along the sections  $\text{Bi}_{5.80}\text{PO}_{11.20}-\text{Nd}_{5.80}\text{PO}_{11.20}$  and  $\text{Bi}_{5.80}\text{PO}_{11.20}-\text{NdPO}_4$  of the  $\text{Bi}_2\text{O}_3-\text{Nd}_2\text{O}_3-\text{P}_2\text{O}_5$  ternary system to get insight into the possibilities for growing neodymium doped  $\text{Bi}_{5.8}\text{PO}_{11.2}$ . Even at low Nd:Bi ratio  $> 0.01$  there remained a small undissolved fraction in the melt, indicating a steep increase of the liquidus temperature. An endothermal effect around  $925^\circ\text{C}$  may be explained by a ternary eutectic in the  $\text{Bi}_2\text{O}_3-\text{Nd}_2\text{O}_3-\text{P}_2\text{O}_5$  system. At concentrations Nd:Bi  $\leq 0.01$  crystal growth experiments were successful with both  $\text{Nd}_{5.8}\text{PO}_{11.2}$  and  $\text{NdPO}_4$  as dopants.

#### CONCLUSIONS

The melting diagram  $\text{Bi}_2\text{O}_3-\text{P}_2\text{O}_5$  [1] has to be revised, there exists no compound  $\text{Bi}_5\text{PO}_{10}$ , instead a non-stoichiometric compound  $\text{Bi}_{5.80}\text{PO}_{11.20}$ , melting congruently at  $951^\circ\text{C}$ , was established. Single crystals of  $\text{Bi}_{5.80}\text{PO}_{11.20}$  can be grown from a melt of this composition, supporting the first conclusion.

$\text{Bi}_{5.80}\text{PO}_{11.20}$  can be doped with neodymium, probably by crystallochemical substitution  $\text{Bi}_{5.8(1-x)}\text{Nd}_{5.8x}\text{PO}_{11.2}$  ( $x \leq 0.01$ ). DTA and thermomicroscopy are useful techniques to be applied in conjunction with single crystal growth.

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